# Synthesis of Zinc Oxide nanoparticles at different aging time for low cost Dye Sensitized Solar Cells

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### Abstract

Zinc oxide nanoparticles synthesized via simple precipitation method and aged at different time were used as photoanodes of dye sensitized solar cells (DSSCs). The structural and the optical properties of the obtained zinc oxide nanoparticles were investigated. The performance of the assembled solar cells was studied and the efficiency of such cell was compared as a relation of the aging time of the ZnO nanoparticles.

#### Key words

Zinc oxide, nanoparticles, precipitation method, dye sensitized solar cell

#### Introduction

The wide band gap semiconductor of band gap > 3.2eV, with a high exciton binding energy of 60 meV and its high electron mobility make Zinc oxide as one of the most interested candidates for many technological multidisciplinary applications, such as in solar energy conversion, thin film transistors, catalysis, nonlinear optics, gas sensors, pigments, cosmetic, etc. (Chou, et al. 2007; Gao and Wang 2005; Hachigo, et al. 1994; Han and Kim 1998; Harada, et al. 1992; Hotchandani and Kamat 1992; Ko, et al. 2003; Lee, et al. 1995; Lieri, et al. 2005; Morkoc, et al. 1994; Sakohara, et al. 1992; Shishiyanu, et al. 2005). Nanostructured size zinc oxide can be found in different forms, like nanoparticles, nanorods, nanobelt, nanoflowers, etc.(Kharisov 2008; Li, et al. 2011; Liu, et al. 2004; Yi, et al. 2005; Yum, et al. 2008). It is favorable for electron transport, easily crystallized and anisotropically grow in a variety of morphologies. Due to its unique properties, it has been proposed as alternative photoelectrodes in Dye Sensitized Solar Cells (DSSCs) in order to achieve better performance.

DSSCs have attracted much attention due to their low cost manufacturing, fabrication on flexible substrate and simple construction (Grätzel 2003; Hagfeldt, et al. 2010). Despite these advantages, the maximum certified efficiency for DSSCs is much lower compared to silicon based solar cells. Much research has been focused to enhance the performance of DSSCs with different anode materials. Common photoanode materials in DSSCs are  $TiO_2$ , ZnO NPs film on transparent conducting oxide (TCO) layers (Grätzel 2003). The popularity of such materials is due to their large surface area and chemical affinity for dye adsorption as well as their suitable energy band potential for charge transfer with dye and electrolytes (Grätzel 1991)(Bandara, et al. 2005). The structure and morphology of the photoelectrode play a vital role in developing such materials (Kuang, et al. 2007). The improvement of the conversion efficiency can be attributed to the larger surface area, which is a goal of most researchers.

In this work, zinc oxide nanoparticles were synthesized using simple precipitation method and aged at different periods. The structural and optical properties of the obtained nanoparticles were characterized using different techniques. Solar cells were assembled using the different synthesized nanoparticles as photoanode. Their performance parameters were investigated and a comparative study was presented.

## **Experimental Set Up**

### Synthesis of the ZnO nanoparticles

ZnO NPs were synthesized using simple precipitation method at different aging time. 8.0 g of  $Zn(CH_3COO)_2 2H_2O$  (Zn Ac) was dissolved into 100 ml of pure methanol under stirring for 30 min at 60°C followed by sonication for 30 min until a clear and vigorous transparence solution was obtained. Dissolving of Zn Ac into hot methanol completely under vigorous storing means total hydrolysis of Zn Ac forming acetate ions and Zn ions. 5M *NaOH* solution was added drop wise into the acetate solution under vigorous stirring until the pH reached 12. Then, the clear solution was transferred to milky white slurry suspension. The white precipitate was stirred for different aging times of 20, 30, 60 and 90 min that were titled as S1, S2, S3 and S4, respectively. After that, the precipitate was stirred slowly at room temperature overnight. White precipitates were carefully collected and washed with absolute ethanol five times for 1.5 hours using centrifugation at 3500 rpm to remove the non reacted materials. Zinc hydroxide can be, easily, converted to ZnO NPs by drying. So, the product dried at 65°C overnight. Finally, the product crashed using mortar and pestle to produce fine powder. The structural properties of the ZnO NPs were characterized using different techniques.

The structural characterization of the obtained nanoparticles was carried out using an X-ray diffractometer (*XRD*) (type: Philips Expert), using Cu anode material operating at 40 kV and 30 mA with ( $\lambda = 1.5418 A^{o}$ ) employing a scanning rate of  $0.02^{o}s^{-1}$ .

Transmission electron microscopy (*TEM*) is a good tool to determine the particle size and morphology of the nanoparticles down to a nanometer scale as well their crystalline structure. Their structure were characterized using a high resolution transmission electron microscopy (*type*: (*JEM* – 2100), *JEOL*).

The UV - VIS absorption measurements have been performed using double beam *Shimadzu UV* – 1601 *PC* with a diffraction gritting with self-aligning, energy- optimizing deuterium lamp for ultraviolet region and halogen lamp for visible region to enhance the accuracy across the UV - VIS spectrum and into the near infrared (*NIR*).

#### Fabricate and assembly the device (Solar cell)

The pastes of *ZnO NPs* were prepared by mixing 0.3 g of *ZnO* nanopowder with 0.6 g of Polyethylene glycol (*PEG*400) using a mortar and pestle for 10 minutes. The pastes were deposited on fluorine-doped SnO<sub>2</sub>(*FTO*) coated glass substrates, using the "doctor blade" technique forming  $0.5 \times 0.5$  cm layers. The layers were dried at 100°C for 60 min followed by sintering them at 450°C for 1 h in air. The obtained sintered layers had a thickness between 12 to  $15 \mu m$ . The prepared layers of *ZnO* photoanode were immersed to sensitize them in 0.32 *mMEosin B* for 24 h. After that films were rinsed in ethanol and were dried in air at room temperature. A Pt-coated *FTO* substrate was used as the counter electrode and an iodide-based solution, consisting of 0.6M tetra - butylammonium iodide, 0.1M iodine and 0.5M 4 - tert - butylpyridine in *acetonitrile* was used as the liquid electrolyte.

Photovoltaic properties of *DSSCs* were characterized using simulated *AM* 1.5 sunlight illumination with an output power of  $100 \ mW/cm^2$ .

### **Results & Discussions**

X-ray diffraction (*XRD*) patterns was used to provide the information about the crystal structure and the grain size of the NPs. Fig. 1 depicts the *XRD* patterns of the of thesamples S1, S2, S3 and S4 of the synthesized *ZnO NPs*. The *XRD* patterns of the synthesized samples matched with the standard hexagonal wurtzite structure phase of *ZnO* (*ICSD card No*.067454). Fig. 1 indicates that the *XRD* patterns have gradual sharpness

of the peaks as the aging time increases, which means that the particle size increases gradually. No other phases have been observed, which further confirmed the formation of pure ZnO NPs with single phase. The sharpness of the broaden XRD diffraction peaks are indicated to the small size and the good crystallinity of the NPs. Different lattice parameters were calculated from XRD patterns of the ZnO NPs and tabulated in table 1, which was agreed with the results were reported in ref. (Thomas 2014). The average particle size (D) of the NPs were estimated from the broadening of the highest intensity peak (101) by Debye-Scherrer's formula, which showed increasing from 18.70 to 26.83 nm with increasing of the aging time from 20 to 90 min. as shown in table 1. The dislocation density ( $\delta = 1/D^2$ ), which represents the amount of defects in the sample is defined as the length of dislocation lines per unit volume (V) of the crystal and irregularity within a crystal structure. (Thomas 2014). The calculated values of  $\delta$  decreased from 2.86 to 1.50  $nm^{-2}$  with the increasing of the particle size and the aging time as depicted in Table 1. The strain ( $\varepsilon = \beta/4 \tan(\theta)$ ) where  $\beta$  is the full width at half maximum), which can be defined by the change in size to the original size of the particle (Ahmad Monsh 2012). The obtained values of  $\varepsilon$  decreased from  $10.3 \times 10^{-4}$  to  $7.46 \times 10^{-4}$  as the particle size increased with increases of the aging time.



Fig. 1: XRD patterns of the samples S1, S2, S3 and S4 of the synthesized ZnO NPs

#	а	С	( <b>c</b> / <b>a</b> )	V	β	D	$\delta * 10^{-3}$	$\epsilon * 10^{-4}$
	( <b>nm</b> )	( <b>nm</b> )		( <b>nm</b> <sup>3</sup> )	( <b>deg</b> .)	( <b>nm</b> )	$(nm^{-2})$	
<b>S1</b>	3.2405	5.1663	1.5943	46.981 6	0.742	18.700	2.86	10.3
<b>S2</b>	3.2373	5.213	1.6102	47.31 20	0.7109	19.504	2.63	9.85
<b>S</b> 3	3.2435	5.2249	1.6109	47.602	0.52	25.781	1.50	7.16
<b>S4</b>	3.241	5.293	1.633	48.150 7	0.500	26.830	1.50	7.46

Table 1: XRD calculated Geometric parameters of the samples S1, S2, S3 and S4 of the synthesized ZnO NPs.

#### High Resolution Transmission Electron Microscopy (HRTEM)

The morphology of the sample S3 of the synthesized *ZnO NPs* was studied by high-resolution transmission electron microscopy (*HRTEM*).

Fig. 2(a) depicts the high-resolution image of sample S3 of *ZnO NPs* and the sharp boundary of the spherical particle can be seen. Fig. 2 (b)shows the diffraction patterns. Theselected electron diffraction area (*SAED*) pattern and the histogram of the sample S3 of the synthesized *ZnO NPs* are shown in Figs. 2(c and d). As shown in Fig. 2(a) *ZnO NPs* was crystalline, spherical and uniform with average particle size 23 nm which is very closed to the estimated averaged particle size of range of 18.7 to 26.8 nm from the *XRD* pattern broadening. The spacing between the adjacent lattice planes distance (d) of the sample S3 of the synthesized *ZnO NPs* was measured from the bright field *SAED* patterns, which was founded to be 0.2457. The obtained value of d of 0.2457 nm is in good agreement with the obtained value from *XRD* results, which corresponds to the (102) plane of *ZnO*.



Fig. 2 HRTEM micrograph of (S3) ZnO nanoparticles sample (a) micrograph image, (b) Diffraction patterns and (c) selected electron diffraction area and (d) histograms

#### **UV-VIS** absorption characterization

The effect of the aging time on the optical properties of synthesized of *ZnO NPs* were studied by UV - VIS absorption spectroscopy. Fig. (3) represents the typical UV - VIS absorption spectra of the samples S1, S2, S3 and S4 of the synthesized ZnO NPs, which are recorded at room temperature. This figure indicates that, the corresponding absorption edge was positioned at 353, 364, 365 and 368 nm for the samples S1, S2, S3 and S4, respectively. These peaks indicates that the absorption edges shifted towards the higher value of wavelength (red shift) with increasing of the aging time. This shift may be attributed to the increasing of the average particle size, which agrees with the study in ref. (Almoqayyad 2012). Based on the UV - VIS absorption spectroscopy of the samples S1, S2, S3 and S4 of the synthesized *ZnO NPs*. The energy gap ( $E_g$ ) was estimated using Tauc Davis and Mott relation ( $\alpha hv$ )<sup>2</sup>  $\propto$  ( $hv - E_g$ ), as shown in Fig. (4). The estimated values of Eg were decreased from 3.52 eV to 3.36 eV as increases of the aging time from 20 to 90 min. This shift may attributed to the increasing of the average particle size, which agrees with studies by (Almoqayyad 2012).



Fig. (3) UV-VIS. spectra of pure ZnO nanoparticle samples S1, S2, S3 and S4 at different aging times using simple precipitation method.



Fig.  $(4)(\alpha h\nu)^2$  versus the photon energy  $(h\nu)$  of pure ZnO nanoparticles S1, S2, S3 and S4 at different aging times using simple precipitation method.

#### **Cell Performance**

The performance of the synthesized *ZnO* as a semiconducting photoanode layer was tested using *EosinB* as dye sensitizer. The characteristics J - V curves of different DSSCs fabricated with different nanoparticles that aged at different times S1 to S4 are depicted in Fig. (5). For each cell, the measured current density versus the applied voltage V is obtained at the incident light intensity of 100 mWm<sup>-2.</sup> The resulted power is plotted versus the applied voltage as shown in Fig. (6). This study shows the benefits of increasing the aging time. The performance parameters of the measurements are summarized in table 2. It is shown that the cell S3 exhibited the highest efficiency and the best performance parameters were obtained. It has an open circuit voltage,  $V_{oc} = 0.374$  V, a photocurrent density  $J_{sc} = 0.890$  mAcm<sup>-2</sup>, maximum power  $P_m = 0.204$  mWcm<sup>-2</sup>, filling factor FF = 61.7 %, and an overall photo to electric conversion efficiency  $\eta = 0.205$  %. The open circuit voltage ranges between 0.34 V for the cell S2 and 0.454 V for S1, while

the current density has the maximum value of 0.985 for the cell S1, and the minimum value of 0.577 for the cell S4. The increase of the open circuit may be attributed to less charge recombination, and the increase of the charge injection results to an increase of the photocurrent density. This may be attributed to more dye loading on the surface of the nanoparticles. The fill factor of the fabricated cells ranges from 49% to 61.7% for S1 and S3, respectively. The highest fill factor was observed for the S3 and the lowest fill factor was obtained for the cell S2.

In general the cell S1and the cell S3, have relatively higher value of the output power and efficiency, on the other hand the cell S2and the cell S4, have relatively lower value of the output power and efficiency. Generally, the low values of current density may be attributed to missing of the anchoring group in the chemical structure of *Eosin B*. Ideal sensitizer must have at least one anchoring group such as,  $-PO_3H_2$ , -Py, -COOHand, $-SO_3H$ . This groups makes a strong link between *ZnO* semiconductor surface and the dye molecule, in order to enhance electron injection from the *LUMO* energy level of the dye molecule to the conduction band of the semiconductor layer (Calogero, et al. 2015).



Fig. (5) J-V curves of pure ZnO nanoparticle samples S1, S2, S3 and S4 aged at different aging times using simple precipitation method.



Fig. (6) P-V curves of pure ZnO nanoparticle samples S1, S2, S3 and S4 at different aging times using simple precipitation method.

**Table2:** Photovoltaic parameters of the DSSCs fabricated using pure ZnO NPs samples S1, S2, S3 and S4 using simple precipitation method.

	J <sub>sc</sub> (mA/cm <sup>2</sup> )	V <sub>oc</sub> (V)	J <sub>m</sub> (mA/cm <sup>2</sup> )	V <sub>m</sub> (mV)	P <sub>m</sub> (mW/cm <sup>2</sup> )	FF %	<b>η</b> %
<b>S1</b>	0.964	0.365	0.763	0.232	0.177	50.30	0.177
S2	0.61	0.34	0.474	0.232	0.11	35.02	0.110
<b>S</b> 3	0.890	0.374	0.768	0.267	0.204	61.70	0.205
<b>S4</b>	0.577	0.454	0.452	0.278	0.127	48.00	0.126

# Conclusion

It was concluded that the aging time of the zinc oxide nanoparticles influences their structural properties. An increase of the particle size was calculated by increasing the aging time, which is in agreement of the shift found in their absorption spectrum. The performance of the solar cells, i.e. the short circuit photocurrent density and the open circuit voltage was affected by the aging time of the obtained nanoparticles. The best photo to electric conversion efficiency for the cell based on the zinc oxide nanoparticles sample (S3) that aged for 60 minutes was the optimum using the low cost *Eosin B* dye with efficiency of 0.205%.

## Acknowledgement

The research activities in Gaza were financially supported by Qatar Charity IBHATH Project grant funded by the Gulf Cooperation Council for the Reconstruction of Gaza through the Islamic Development Bank.

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# تخليق جسيمات نانوية من أكسيد الزنك عند أزمنة تسخين مختلفة باستخدام طريقة الترسيب البسيطة

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### ملخص العربى

في هذا البحث تم تخليق جسيمات نانوية من أكسيد الزنك باستخدام طريقة الترسيب البسيطة و عند أزمنة تسخين مختلة لانتاج خلايا شمسية صبغية رخيصة الثمن. وقد تم دراسة بعض الخواص التركييية والضوئية للعينات. وقيست منحنيات الخصائص الكهربية المميزة لكل خلية مصنعة كعلاقة بين تغير التيار مع الفولت. تم تعينن البار اميتيرات الهامة لكل خلية مثل تيار القصر  $J_{sc}$  و جهد الدائرة المفتوحة  $V_{oc}$  و اعلي قدرة كهربية للخلية  $\eta$