# Evaluation of the Microstructure, Micro Hardness, Electrical Resistivity and Ultrasonic Properties of 6201Alloy Containing Lanthanum

\*Amira Nassar<sup>1</sup> and L. Abd El-Latif<sup>2</sup>

<sup>1</sup>Solid State physics Department, National research Centre, 33 El Bohouth St.12622 Dokki, Giza, Egypt. <sup>2</sup>Ultrasonic Laboratory, National Institute for Standards, Tersa Street, P.O. Box 136, El-Haram, El-Giza 12211, Egypt. \* E-mail <u>fatmohmar@yahoo.com</u>

This research is to investigate the influence of adding different concentrations of (0.0109, 0.0185 and 0.037wt %) La on the microstructure, electrical resistivity, Vickers micro hardness (Hv) and ultrasonic properties on 6201 aluminum alloy comparing to (6201alloy +0.021wt%TiB) used in an electrical industry. The results showed that the (6201alloy+0.037wt% La) was suitable materials for using electrical conductor than alloys added with0.0185wt%La and 0.0109wt% La. It was also found that Vickers micro hardness (Hv) measured mechanically and electrical resistivity for the alloys contaning La has the same trend to those determined from ultrasonic calculations. It was also found that the structural changes inside the matrix of the alloys play an important role in the ultrasonic wave propagation and also in controlling the electrical resistivity and the alloys strength.

# 1. Introduction

In electrical applications, the commercially pure aluminum possesses the highest electrical conductivity among aluminum conductors. Its application is only restricted due to its low strength [1]. Series of aluminum alloys containing Mg and Si as major addition elements show a high strength-to- density ratio, a good electrical conductivity ability and a superior corrosion resistance. In recent years, the aluminum alloy wires made from AA6201 with high tensile strength and medium electrical conductivity are widely used in electrical engineering field. The common disadvantage of the aluminum alloy wires made from AA6201 is "the higher tensile strength, the lower electrical conductivity" [2, 3]. In these alloys, Mg and Si combine to form the chemical compound  $Mg_2Si$  (magnesium-silicide), which is considered as the primary hardening phase [4, 5].

Grain refining is one of the predominant techniques in controlling the strength and toughness and also for subsequent secondary processing of castings. Grain refinement by addition of grain-refiners like titanium boride referred to as inoculation [6] is the most popular due to its simplicity. Sometimes a master alloy of Al-TiB is made and used subsequently to grain refining.

In recent studies, the rare earth La was added into the Al-alloy to investigate its effect on the Al. However the additional La can improve the welding properties of aluminum alloy (Al-Mg-Si) by promoting the diffusion of Mg, Si in base metal and decreasing conglomeration of impurities in unit area of grain boundary. Apart from these, La is an effective grain refinement additive during the casting process. A proper amount of La could modify the microstructure of the Al alloys and improve its mechanical property, electrical conductivity and thermal-resistance [7, 8].

In a wide range of different ultrasonic nondestructive methods, the evaluation of material properties by ultrasonic measurements is introduced as a quality control method, by which the need for conventional destructive tests would be minimized. It has been proven that indirect assessment of the microstructure by ultrasonic velocity or attenuation measurements is possible. However, ultrasonic is based on a simple principle of physics in which any wave will affect by the medium through which it travels. Thus, changes in the micro structural features which control various mechanical properties can be sensed by ultrasonic measurements. [9,10]. Therefore, establishing a relationship between the micro structural parameters and ultrasonic evaluation results may be useful for fast and nondestructive quality control of the materials.

The main object of this work is to compare the microstructure, electrical resistivity, ultrasonic properties and Vickers micro hardness behavior of 6201alloy in the presence of different concentrations of La and in the presence of TiB which used in an electrical industry.

### 2. Experimental

The base metal 99.6% aluminum was melted in a graphite crucible using heating furnace. The required amounts of (0.6-0.9wt %) Mg and (0.5-0.9wt %) Si was added into the molten to produce 6201 alloy used in electrical industry. Each of the master alloy of Al-TiB and Al-La with different percentages were added to the above melt and stirred with a graphite rod to produce a series of modifying alloys namely (6201+0.021wt%TiB, 6201+0.0185wt%La, 6201+0.037wt%La and 6201+0.0109wt%La). The melts were poured into iron molds to get casting alloys in the form of cylinder (200mm in length and 20mm in diameter). Specimens were cut from the as-cast rod for different measurements.

All the specimens for microstructure characterization were cut from the cast cylinder then polishing and etching by hydrofluoric acid. The micro structural analysis of the samples was carried out by using Environmental Scanning Electron Mcroscope (FEI-S). The phases of the samples were analyzed by the EDS system of SEM and X-Ray Diffractometer (XRD) system EMPYREAN by  $Cuk_{\alpha}$  radiation with a wavelength,  $\lambda$ , of 1.54060A°.

Micro hardness (Hv) measurements are evaluated by using a Vickers micro hardness tester with a load of 1.96 N for 5 s.

For the measurement of the electrical resistivity ( $\rho$ ), the samples were prepared in form of discs 2cm diameter and 0.2cm thickness .The surfaces of the samples were polished and having two parallel faces. The direction of measurement was always at contact surface area of the sample. Resistance (R) measurement was carried out using Hioki 3522-50LCR Hitester (Bridge) at room temperature, then the electrical resistivity( $\rho$ ) of the samples was calculated from the following eq.(1)

$$\rho = \frac{RA}{L} (1)$$

where L is the thickness of the sample in cm, and A is the surface area of the sample in  $cm^2$ .

For ultrasonic measurement, discs with dimensions of 10mm x15mm x 15mm were polished to obtain smooth and uniform surface.

The density (D) of the samples was measured by applying Archimedes principle at room temperature using toluene as an immersion liquid according to the formula:

$$\mathbf{D} = \mathbf{D}_{\ell} \left[ W_{\mathfrak{a}}'(W_{\mathfrak{a}+}W_{\ell}) \right] \tag{2}$$

Where  $D_{\ell}$  is the density of toluene,  $W_a$  and  $W_{\ell}$  are the sample weights in air and liquid, respectively.

The ultrasonic wave velocity measurements were carried out by applying the pulse-echo technique. The elapsed time between the initiation and the receipt of the pulse appearing on the screen of a flaw detector (GE model: USN60) was determined. The ultrasonic wave velocity was therefore, calculated by dividing the round trip distance (twice the thickness of the sample) by the elapsed time according to equation (3);

$$V=2X/\Delta t \tag{3}$$

where X is the sample thickness and  $\Delta t$  is the time interval.

All ultrasonic wave velocity measurements were carried out at room temperature using Karl Deutsch transducer S24 HB4 with fundamental frequency of 4MHz to measure longitudinal ultrasonic wave velocity, and Krautkramer transducer K2NY with fundamental frequency of 2MHz for measuring shear ultrasonic wave velocity. The estimated uncertainty (Type A) in ultrasonic wave velocity measurements was  $\pm$  40m/s for longitudinal ultrasonic wave velocity (V<sub>L</sub>) and  $\pm$ 15m/s for shear ultrasonic wave velocity (Vs).

Young's modulus (E), shear modulus (G), longitudinal modulus (L), bulk modulus (K), microhardness (H) and Poisson's ratio ( $\sigma$ ) of the investigated samples have been determined from the measured ultrasonic wave velocities and density measurement as equations: [11,12].

(4)
(5)
(6)
(7)
(8)
(9)

# 3. Results and Discussion

### 3. 1. Microstructure characteristic

The microstructures of 6201 alloy with TiB and La addition in various proportions (i.e, 0.021wt%TiB, 0.0185wt%La, 0.037 wt% La and 0.0109 wt% La) were characterized by SEM as shown in Figs.(1 a, b, c and d), respectively. It is evident from these figures that grains are fine, spherical and rod-like shaped within the grain body and along grain boundaries.



Fig.(1): SEM metallography of 6201 alloy modified with TiB and La addition in various proportions:
(a) 0.021wt%TiB<sub>2</sub>,
(b) 0.0185wt%La,
(c) 0.037wt%La, and
(d) 0.0109wt%La respectively.

XRD and EDS were then performed to identify the present phase and their chemical compositions are shown in Figs. (2 and 3) respectively. Fig. (2) displayed the resulting diffraction pattern recorded from the (6201alloy+0.021 wt% TiB) and (6201 alloy+0.0185wt%La). Due to the low content of Mg, Si, La and TiB in these alloys, it was not possible to observe the compositional analyzes of various phases, as reported in previous results [13]. It was found that the phase present in all diffraction patterns was Al (Reference Code: 01-072-3440). According to the results of EDS show the Al, Mg and La elements in the (6201 alloy+0.037wt% La), while Al and Mg are observed in (6201alloy+0.021wt%TiB) as explained before [13].



Fig. (2): X-ray diffraction patterns of (a) 6201alloy+0.021wt%TiB and (b) 6201 alloy+0.0185wt%La

#### 3. 2. Analysis of Mechanical, Electrical and Ultrasonic Properties

Experimental values of electrical resistivity,  $\rho$  and Vickers microhardess, Hv for 6201 alloys with addition of TiB and different values of La are listed in Table (1). The results showed that the sample alloyed with addition of 0.037wt%La is suitable materials for using electrical conductor, because its mechanical and electrical resistivity values are approximately closed to (6201 alloy+0.021 wt% TiB) used in electrical industry. However, the formation of fine titanium boride, TiB compound causes decreasing harmful effect of the transition element Ti on the conductivity. On the other hand, the harmful effects of Si on the electrical conductivity of Al-Mg-Si aluminum alloys are reduced by adding 0.037wt%La which could form phases containing La and Si in the matrix [2,8]. This is referred to the addition of an element to another results in the formation of a compound. The effect of this compound on the electrical resistivity of a metal is far less significant than that when the alloying element is in solid solution [14]. It was also

found that 0.0185wt% amount of La addition has high strength with low conductivity. Meanwhile, 0.0109wt% La addition has low strength with high conductivity. However, the electric alresistivity of alloys increases with the amount of elements in solid solution [2,14]. While, the strength and hardness of some metallic alloys may be enhanced by the formation of extremely small uniformly dispersed particles of a second phase within the matrix [4]. Hence, alloying elements and some available impurities in the base metal increase strength of the alloy but cause a decrease in conductivity. Therefore, the problem of high tensile with low electric conductivity is still unsolved [15]. So when selecting an aluminum alloy for using in overhead conductors, it is necessary to achieve a compromise between mechanical performance and electrical conductivity.



Fig. (3): EDS of 6201 alloy with (a) 0.021wt% TiB and (b) 0.037wt% Laudations.

Table	(1): Variation	of electrical	l resistivity (ρ)	) and V	/ickers I	Microhardness	(Hv)
	for 6201 allo	ys with vary	ing contents of	La and	d TiB re	spectively	

Alloy	ρ(Ω.cm)	Hv(N/mm <sup>2</sup> )
0.021TiB	0.141	57.68
0.037La	0.132	55.41
0.0185La	0.317	75.8
0.0109La	0.181	61.82

Density, ultrasonic wave velocities ((longitudinal & shear), elastic module (Young's modulus, shear modulus, longitudinal modulus, bulk modulus), Poisson's ratio and microhardness for 6201 alloys with varying contents of La and

TiB respectively are listed in Table (2). It was found that the density of (6201 alloy+0.0185 La) was to be slightly higher than that of the (6201alloy +0.0109 La) while the density of (6201alloy+ 0.037La) was the smallest density. The densities of precipitates play an important role in the ultrasonic wave, as agreement in previous results [16]. However, the density of precipitated phases is determined by the chemical composition of the alloy, so the ultrasonic wave velocities (longitudinal & shear) values dependence on the density of precipitated phases which are contributed to form in the connectivity network structure.

**Table (2):** Variation of Density (D), Longitudinal( $V_L$ ), Shear ( $V_S$ )ultrasonic velocities, Young's modulus (E), Bulk modulus (K), Poisson's ratio ( $\sigma$ )and Microhardness (H) for 6201 alloys with varying contents of La and TiB respectively

Alloy	$D,g/cm^3$	$V_L(m/s)$	V <sub>s</sub> (m/s)	E(Gpa)	K(Gpa)	σ	H(Gpa)
0.021TiB	2.70	6468	3237	75.418	75.234	0.333	3.152
0.037La	2.69	6074	3003	64.925	66.898	0.3382	2.616
0.0185La	2.696	66557	3243	75.881	78.107	0.338	3.060
0.0109La	2.695	6379	3155	71.79	73.896	0.3381	2.895

## 4. Conclusions

The experimental results indicated clearly an influence of La and TiB on the microstructure, microhardness and ultrasonic parameter of 6201 alloy. However, the balance between the electrical conductivity and microhardness of the investigated 6201 alloy is influenced by both the kind and the amount of additions .It was found that 6201 alloy modifted with 0.037wt% La is suitable materials for using electrical conductor than the other concentrations of La. It was also found that electrical resistivity and Vickers microhardness (Hv) measured mechanically for the 6201 alloy content La had the same trend as those determined from ultrasonically.

### Acknowledgments

The authors would like to express their deepest gratitude to the members and workers in the Egyptian Company of Cables and Egypt alum Compant (Nag-Hammadt) for their facilities and valuable help, Particulary Dr. Ezzat Abd-Elrahman El-Katatny for his technical assistance during the preparation of the alloys.

#### **References**:

- 1. P. Lamarâo, C. Santana, K. Kamizono, I. Jorge, *solidified, Mater, Sci*, **20**, 262 (2014).
- 2. S. Karabay, *Mater Des*, 27, 821 (2006).
- 3. S. Karabay, *Mater Des*, 29, 1364 (2008).
- 4. G. Al-Marahleh, American journal of applied Sciences, 3(5), 1819 (2006).
- 5. W. Liping, G. Erjun, M. Baoxia, J. rare earth, 26, 105 (2008).
- 6. D. P. Mondal, N. Jha, A. Badkul, S. Das, *Nonferrous Met. Soc. China*, 22, 1001 (2012).
- 7. H. Xin, Y. Hong, J. of Wuhan University of Technology-Mater Sci Ed, 28, 202 (2013).
- 8. W. Yuan, Z. Liang, C. Zhang, L. Wei, *Mater Des*, 34, 788 (2012).
- 9. C. H. gür, B. Ogel, Mater Charact, 47, 227 (2001).
- 10. S. H. Tang, J. W. Hao, H. L. Pan, NDT&E International, 40, 486 (2007).
- 11. H.Afifi,S. Marzouk, Mater Chem Phys, 80, 517 (2003).
- 12. F. M. Helaly, S.H. El-Sabbagh, L. Abd-El-latif, *Polym Polym compos*, 13(3), 1 (2005).
- 13. S. DelaChapelle, P. Duval, *Texture Microstruct*, 35(2), 55 (2002).
- 14. P. L. Mangonon, Ph. D., P. E., FASM, (Florida Institute Techology Melbourne) P.93 (1999).
- **15.** T.G. Zhou, Z.Y. Jiang, J. L. Wen, H. Li, A. K. Tieu, *Mater Sci, Eng*, A, **485**, 108 (2008).
- 16. C. H. Gür, I. Yildiz, J. Non destruct Eval, 27, 99 (2008).