Structural and electrical properties of Sprayed ZnO:Al thin films

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Abstract

Aluminium doped zinc oxide (AZO) thin films were prepared on glass substrates by spray pyrolysis technique. The Al concentration in the starting solution was varied from (0 to 5) wt %. XRD revealed that both pure and AZO thin films were polycrystalline with hexagonal structure and exhibited (002) preferential orientation. AZO film with minimum electrical resistivity $8.61 \times 10^{-3} \Omega$ cm were obtained at a doping ratio of 2 wt. %.

Keywords: Spray pyrolysis; ZnO; structural and Electrical properties.

1. Introduction

Thin films of non-stoichiometric and doped metallic oxides of ZnO, SnO₂ and ZnO: Al etc. had attracted a large attention because of their electrical and optical properties making them suitable for application in different kinds of opto-electronic devices (**Rozati. S.M, etal 2007**). Among these materials, ZnO thin films have recently gained much attention due to its advantages over other oxide thin films like ITO, SnO₂, etc. These advantages include non- toxicity, low cost, high electrical conductivity and transparency (**Zhou.H, etal 2007**), ZnO is a n-type semiconductor with an optical band gap about 3.3 eV consequently; it exhibits a high transmittance in the visible region (**Nunes.P, etal 1999**). Stoichiometric pure zinc oxide is an insulator while its conductivity could be enhanced by adding Group III metal dopants, such as Al, In and Ga or Group VII dopants like F (**Ellmer Klaus, etal 2008**).

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Both Al and Zn are low cost, non-toxic elements with abundant supply, which make ZnO:Al more suitable than indium tin oxide as TCO layer in photovoltaic applications. There are several deposition techniques that have been employed to grow ZnO thin films like chemical vapor deposition (CVD) (Haga.K, etal 1999), magnetron sputtering (Minami.T, etal 2000), pulsed laser deposition (PLD) (Hyuck Bae.S, etal 2000), sol–gel process (Lee.G.H, etal 2003) and spray pyrolysis (SP) (Krunks.M, etal 1995, Paraguay.F, etal 1999). Among these methods, the spray pyrolysis technique has several advantages such as simplicity, safety and the low cost of the apparatus and the raw materials (Jin-Hong Lee, etal 2004). In this paper we investigate the effects of Al doping concentration on the structural and electrical properties of ZnO thin films.

2. Materials and Methods

The pure and AZO thin films were deposited onto glass substrates by using home made spraying equipment at deposition temperature 450 °C .The Al concentration in the starting solution was varied from (0 to 5) wt. % keeping all other parameter constant. The spraying solution was 0.3 M zinc acetate dehydrate (Zn (CH₃COO)₂.2H₂O) and aluminium chloride (AlCl₃) dissolved in distilled water mixed with ethanol (C₂H₅OH) in the ratio of (1:3), to stabilize the starting solution a few droplets of acetic acid (CH₃COOH) 96% was added. The X-ray diffraction (XRD) of prepared films was obtained by Philips diffractometer model (PW3040) using K α_1 radiation (λ =1.54056 A°) in range 2 θ = (30°–70°). The surface morphology of the films was examined by Atomic force microscopy (AFM) Shimadzu Contact mode Model Wet–SPM. The films composition energy dispersive X-ray analysis (EDX) was examined by JEOL (JXA_840A) electron prop-micron-alyzer. The electric properties of deposited films were characterized by Hall effect measurement Lake Shore model -7700A.

3. Results and Discussion

The XRD patterns of AZO thin films deposited at various Al ratios (0 to 5) wt. % are shown in Fig.1. The XRD analysis revealed that all deposited films are polycrystalline with hexagonal wurtzite structure. The intensity of the (002) peak decreases as the concentration

of aluminium increases which is possibly due to segregation of Al_2O_3 into the grain boundaries ,which inhibits the growth of the crystallite size in the films as the Al concentration increases (**Nunesa.P, etal 2002**). The films with Al concentration less than 2 wt. % shows the most predominant peak is (002) peak where the (101), (102) and (103) peaks are smaller compared to (002) peak indicating preferred orientation along the c-axis perpendicular to the substrate surface. Films doped with Al more than 2 wt. %, shows decreasing in its crystallinity and the intensity of (002) peak become comparable with (101) peak and it was hard to detect (102) and (103) peaks. This deterioration of the films crystallinity may be explain due to the formation of amorphous Al_2O_3 phase and/or segregation of aluminumin grain boundaries (**Dghoughi.L, etal 2010**). A similar trend was observed by EI Manounia et al. (**EI Manounia.A, etal 2006**) and Nunesa et al. (**Nunesa.P, etal 2002**) for the doping range of (1-4) at.% and (1– 5) at.%.



Fig.1 XRD patterns of AZO films with various aluminium concentrations.

The average crystallite size has been calculated using Deby -Scherrer formula (Ohta.Y, etal 1997);

$$D = \frac{0.9\lambda}{\beta\cos\theta} \tag{1}$$

Where β is the observed angular width at half maximum intensity (FWHM) of the peak, λ is the X-ray wavelength (1.54056 A° K_{α 1} radiation) and θ is the Bragg's angle.



Fig.2 Crystallite size of AZO films with various aluminium concentrations. Fig.2 shows the variation in the crystallite sizes as function of Al concentration. As it seen the crystallite sizes decreased with the aluminium concentration increase from 65.98 nm to 9.55 nm. Which may be explain due to the segregation of Al_2O_3 into the grain boundaries which inhibits the growth of the particle size in the films as the Al concentration increases (**Muiva.C.M, etal 2011**).



Fig.3 AFM images of ZnO thin film: [a] pure ZnO and [b] AZO thin film 2%.wt

The typical AFM images of pure and AZO thin films with Al concentration (2%.wt) are shown in Fig.3. The scanning areas are 5µm×5µm. ZnO film exhibit columnar morphology with smooth surface compared to rough surface of AZO film. A similar behavior was observed by J.J. Ding (**Ding.J.J ,etal 2009**).

The films composition was examined by Energy Dispersive X-ray Spectroscopy (EDX). This study shows the expected elements: Zn, O and Al. The calculated weight concentrations for the elements are presented in Table (1).

Al (%)	Zn (%)	O (%)	Al (%)	O/Zn	Al/Zn
0	52.44	47.56	0	0.91	0
1	53.78	45.69	0.53	0.85	0.01
2	46.24	52.55	1.21	1.136	0.0262
3	42.50	55.24	2.27	1.299	0.0534
4	43.60	53.0	3.40	1.216	0.078
5	42.87	52.9	4.22	1.234	0.099

Table (1): EDX analysis of AZO thin films deposited with various aluminium concentration.

For pure ZnO film the [O]/[Zn] ratio is smaller than unity indicating a slight oxygen deficiency, which acts as donors leading to n-type conductivity, also shows relatively the best stoichiometrie ([O]/[Zn]=0.91) .A Al doping concentration increase in the prepared solution, the [Zn] concentrations decrease and [Al] concentrations increase causing that [Al]/[Zn] ratio increase which may be considered as prove that Zn⁻² ions is replacement by Al⁻³ ions leading to n-type conductivity and improvement in its electrical properties (**DghoughiL, etal 2010**).

The electrical properties of AZO films prepared with different Al concentration (0-5) wt.% were measured and summarized in Table (2). It can be observed that the resistivity decreased from 1.14 Ω . cm for pure ZnO to a minimum of $8.61 \times 10^{-3} \Omega$.cm for films prepared with a starting solution containing 2 wt.%. The resistivity then started increasing with doping concentration to 1.11 Ω . cm at 5 wt.% doping. This shows that the resistivity reaches a minimum at 2 wt.% Al doping. This result is quite comparable to other reports (**Muiva.C.M, etal 2011**).

Al concentration	Resistivity	Carrier	Mobility	Conductivity
(%)	$(\Omega.cm)$	concentration	(cm²/vs)	type
		(cm ⁻³)		
0	1.14	3.3×10^{17}	16.6	n
1	6.54×10^{-2}	4.5×10^{18}	21.2	n
2	8.61×10 ⁻³	3.14×10^{19}	23.1	n
3	3.05×10^{-2}	1.13×10^{19}	18.1	n
4	8.69×10^{-1}	3.34×10^{18}	2.15	n
5	1.11	2.62×10^{18}	2.16	n

Table (2): Electrical parameter measurements of AZO films prepared at various Al.

When a small amount of Al was introduced into the ZnO film, the Al was ionized into Al^{3+} and substituted for Zn^{2+} . Thus, one free electron was produced from one zinc atom replaced by an aluminium atom. Small amounts of aluminium introduce large numbers of free electrons in the doped films, and the conductivity therefore increases. But further increase of aluminium concentration (>2 wt.%) does not further increase the conductivity. Excess amount of aluminium cannot be accommodated into ZnO lattice due to its limited solid solubility and therefore form neutral aluminium oxide and segregates in the grain boundaries (**Gomez-Pozos.H, etal 2007**). Therefore, the amount of electrically active Al atoms in the film is reduced when the Al doping is high and excess dopant lead to build up of carrier traps in the lattice reducing their mobility. In the case of the films deposited with the optimum Al ratio (2 wt.%), we suppose the solubility limit of Al into ZnO lattice is reached hence Al is uniformly distributed inside the grains and a continuous film is obtained, and consequently the electrical resistivity reaches its minimum value (**Muiva.C.M, etal 2011**).

4. Conclusion

Highly transparent conducting AZO thin films were successfully prepared on glass substrate by spray pyrolysis technique. The influences of Al doping concentration on structure and electrical properties were studied. All prepared films were polycrystalline with hexagonal wurtzite structure. The film obtained with Al concentration 2 wt. % show best crystallinity and minimum resistivity $8.61 \times 10^{-3} \Omega$.cm.

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الملخص باللغة العربية

الخواص التركبية و الكهربائية للأغشية الرقيقة من أكسيد الزنك المطعم بالالومنيوم

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تم تحضير عينات من أكسيد الزنك النقى و المطعم بالالومنيوم بطريقة الأحلال الحرارى للرذاذ. و قد تم دراسة تأثير تركيز الالومنيوم على كل من الخواص التركبية و الكهربية حيث أظهر تالنتائج أن الأغشية المحضرة متعددة التبلور و أن نسبة التبلور تقل و تضمحل بزيادة نسبة عنصر الالومنيوم. كما أظهرت النتائج أن الالأغشية الرقيقة من أكسيد الزنك المطعم تمتاز بتوصيلية كهربائية عالية من النوع السالب و قد لوحظ أن أفضل خواص كهربائية قد تحققت عند نسبة تطعيم 2 %.